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Thermo-rheological modelling of cement-bitumen treated materials in the small strain domain

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(Article begins on next page)

# 1 **Thermo-Rheological Modelling of Cement-Bitumen Treated** 2 **Materials in the Small Strain Domain**

## 3 4 **Abstract**

5  
6 Cold recycled materials (CRM) have been introduced as structural materials in road pavement  
7 structures thanks to their significant economical and environmental benefits. Among them,  
8 cement-bitumen treated materials (CBTM) are often employed because of both contributions given  
9 by bitumen (in form of emulsion) and cement. The first confers a bituminous behaviour, whereas  
10 the second ensures good short-term performance otherwise penalized by the presence of water.  
11 Water plays a fundamental role in providing workability of the mixture at the atmospheric  
12 production temperatures. Due to such peculiarities, CBTM mixtures require attention when  
13 rheological modelling is performed in the small strain domain. This paper provides an overview  
14 on the most common rheological model applied to bituminous mixtures (2S2P1D) and the main  
15 issues related to the application to CBTM mixtures are highlighted. Afterwards, another model is  
16 proposed from the literature, the DBN model, and applied to three mixtures. The mixtures were  
17 prepared to assess the effect of the bitumen emulsion used, as well as the type of curing conditions.  
18 Results showed that the DBN model seems to be an excellent tool for not only CBTM rheological  
19 modelling in the small strain domain and it is recommended for applications in wider experimental  
20 programs.

21  
22 **Keywords:** Cold recycling, Bitumen emulsion, Reclaimed Asphalt Pavement, Small strain  
23 domain, Rheological modelling

## 26 1 INTRODUCTION

27 The increasing costs related to the road industry to face the necessity of frequent maintenance and  
28 rehabilitation projects have led to the promotion of sustainable technologies characterized by  
29 substantial environmental and economic benefits. Among them, cold recycled materials (CRM)  
30 are the most promising materials for structural layers (base or subbase), due to the possibility of  
31 reusing high quantity of reclaimed asphalt and performing the production process at atmospheric  
32 temperature (ARRA, 2001, 2016; Giani et al., 2015; Lauter & Corbett, 1998; Omrani & Modarres,  
33 2018; Stroup-Gardiner, 2011). In fact, the bitumen is used in form of bitumen emulsion or foamed  
34 bitumen, whereas the workability of the mixtures is ensured by the addition of water (Asphalt  
35 Academy, 2009). A specific type of CRM materials is the cement-bitumen treated materials  
36 (CBTM), where the short and long-term properties are improved by the addition of ordinary  
37 Portland cement. Such materials show a dual behaviour (asphalt-like and cement-like) when  
38 bitumen and cement are employed at almost the same dosages (between 1 and 3%) (Bocci et al.,  
39 2011; Cardone et al., 2014; Chen et al., 2020; Grilli et al., 2012). In addition to the low amount of  
40 bituminous binder and the presence of cement, an important difference between CBTM and the  
41 traditional hot mix asphalts (HMA) is the use of reclaimed asphalt pavement (RAP) as a black  
42 rock (Raschia et al., 2019). The RAP is generally employed in high amount in the CBTM aggregate  
43 gradation and it is reasonable to assume that, at atmospheric temperature, the aged binder coating  
44 the aggregate particles does not blend with the added virgin binder (emulsion or foam).

45 In the literature, mechanical characterization of CBTM mixtures is commonly performed  
46 by means of traditional tests, often empirical, determining the resistance at high deformations  
47 (failure) (Dal Ben & Jenkins, 2014; Graziani et al., 2018; Hodgkinson & Visser, 2004; Kim et al.,  
48 2011; Zhu et al., 2019). Only few feedbacks can be found on CBTM (or CRM) mixtures  
49 characterization in the small strain domain (Chomicz-Kowalska & Maciejewski, 2020; Gandi et  
50 al., 2017; Godenzoni et al., 2015; Godenzoni et al., 2016; Saleh, 2007), even though in many cases  
51 sigmoidal functions characterized by experimental parameters are preferred instead of rheological  
52 models composed by specific elements (springs and/or dashpots) (Graziani et al., 2020). In  
53 particular, the determination and modelling of rheological properties, such as complex modulus  
54 and phase angle, allow a better understanding of the material behaviour by means of the application  
55 of a valid rheological model. In case of bituminous mixtures, the Huet-Sayegh model was widely  
56 used in the past, even though it does not allow to well represent the material behaviour at very low

57 frequencies (or very high temperatures) (Olard & Di Benedetto, 2003; Pronk, 2006). For this  
58 reason, the 2S2P1D was introduced to obtain a complete rheological description of bituminous  
59 mixtures and binders in the linear viscoelastic field (Olard et al., 2003). Of course, the application  
60 of this model is valid in any case where the loading conditions (number of cycles, strain amplitude,  
61 temperature) keep the material in the linear domain. When non-linearities are present, a more  
62 versatile model can be used, like the DBN model (Di Benedetto, Mondher, et al., 2007). In fact,  
63 the DBN model can be applied depending on the strain level and the formulation can be quite  
64 simple (in case of linear viscoelasticity) or more complex (permanent deformation or fatigue).

65 The **main** objective of this paper is to define and employ a suitable rheological model to  
66 investigate the properties of CBTM mixtures in the small strain domain. The most common models  
67 proposed by the literature **are described, highlighting** the main issues observed. **Furthermore, as a**  
68 **preliminary and validation step, the new approach is applied to a limited number of specimens.**

## 69 **2 THERMO-RHEOLOGICAL MODELLING OF HMA AND CBTM**

70 The main characteristic of CBTM mixtures **is the presence of both bitumen and cement as binding**  
71 **agents.** The contribution of both binding agents makes the thermo-mechanical and rheological  
72 description of such materials different from the traditional approaches followed for HMAs. In fact,  
73 CBTM mixtures could be considered as an intermediate material **between bitumen-stabilized**  
74 **mixtures, cement treated mixtures and bituminous mixtures (Grilli et al., 2012).** Rheological  
75 models developed so far are suitable for systems where the dissipation at small strain level can be  
76 explained when considering LVE behaviour. This assumption is considered valid probably  
77 because HMA are characterized by higher effective bitumen content and lower voids when  
78 compared to CBTM mixtures. For these mixtures, the aggregates are not completely coated by the  
79 bitumen film, which is instead dispersed irregularly, and the use of RAP as a black rock implies  
80 the presence of the aged binder in addition to the virgin binder (Asphalt Academy, 2009). Such  
81 considerations can explain, at a local scale, the observed behaviour during rheological testing and  
82 must be taken into account for the rheological modelling of CBTM mixtures.

83 The linear viscoelastic Huet-Sayegh model is a rather good tool to represent the rheological  
84 properties of bituminous mixtures but not binders, especially at low frequencies (or high  
85 temperatures). The analytical expression of the Huet-Sayegh complex modulus is described in Eq.  
86 (1):

87

$$88 \quad E_{HS}^*(i\omega\tau_E) = E_{00} + \frac{E_0 - E_{00}}{1 + \delta(i\omega\tau_E)^{-k} + (i\omega\tau_E)^{-h}} \quad (1)$$

89

90 where  $i$  is the complex number defined as  $i^2 = -1$ ,  $\omega$  is the pulsation defined as  $\omega = 2\pi f$ ,  $f$  is the  
91 frequency,  $k$  and  $h$  are constant exponents ( $0 < k < h < 1$ ),  $\delta$  is a constant,  $E_{00}$  is the static  
92 modulus for  $\omega \rightarrow 0$ ,  $E_0$  is the glassy modulus when  $\omega \rightarrow \infty$ , and  $\tau_E$  is the characteristic time, which  
93 is the only parameter depending on the temperature.

94

$$95 \quad \tau_E(T) = a_T(T) \cdot \tau_{0E} \quad (2)$$

96

97 where  $a_T(T)$  is the shift factor at a temperature  $T$ ,  $\tau_E(T) = \tau_{0E}$  at the reference temperature  $T_0$  and  
98  $\tau_E(T)$  is determined at each isotherm by minimizing the error between the measured and modelled  
99 norm of the complex modulus,  $|E^*|$ .

100

101 It was recently observed that applying the Huet-Sayegh model to CBTM and focusing the  
102 fitting on the  $|E^*|$  led to a systematic error in the modelling of the phase angle ( $\varphi$ ), parameter  
103 characterizing the viscous energy dissipation (Graziani et al., 2020). In particular, a constant phase  
104 lag independent of temperature and frequency was observed between the modelled and  
105 experimental values, stressing the fact that the model underestimates the total energy dissipation  
106 considering only the viscous component. As a result, it seemed that CBTM mixtures are  
107 characterized by a total energy dissipation composed of a viscous component and non-linear  
108 phenomena (non-viscous). The authors described this aspect as energy dissipation probably due to  
109 the aggregate-to-aggregate contact and friction. As a solution, they proposed an analytical  
110 modification to the Huet-Sayegh equation, which consists in the addition of a constant phase angle  
111 and expressed by Eq. (3).

111

$$112 \quad E_{HSq}^*(i\omega\tau_E) = E_{HS}^*(i\omega\tau_E) \cdot \exp\left(iq \frac{\pi}{2}\right) \quad (3)$$

113

114 where  $E_{HS}^*(i\omega\tau_E)$  is the Huet-Sayegh model (Eq. (1)) and the term  $\exp\left(iq\frac{\pi}{2}\right)$  represents an  
115 additional dissipation element with an angle,  $\varphi_{AEP}$ , equal to  $q\frac{\pi}{2}$  without affecting the absolute  
116 value of the complex modulus.

117 Such correction led to a better fitting of the experimental data obtained for CBTM mixtures,  
118 but in this form, the model is only suitable for sinusoidal loading and cannot be used to extend the  
119 material representation in the time domain for another loading path. This drawback does not exist  
120 for the DBN model presented below. In the literature, the 2S2P1D model is extensively used to  
121 describe the rheological behaviour of bituminous mixtures in the LVE field with good  
122 approximation. In addition, the parameters that define the 2S2P1D model are used in the  
123 calibration process of the DBN model.

## 124 **2.1 2S2P1D model**

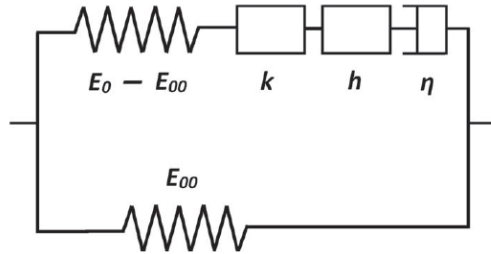
125 The 2S2P1D model is a linear viscoelastic rheological model composed of 2 springs, 2 parabolic  
126 elements and 1 dashpot. In particular, one spring is placed in parallel with a series of the remaining  
127 elements (Figure 1). Thanks to its nature, this model is largely employed to model unidimensional  
128 or tridimensional behaviour of bituminous materials (binders, mastics and mixtures) (Di  
129 Benedetto, Delaporte, et al., 2007; Di Benedetto et al., 2004; Tiouajni et al., 2011). The analytical  
130 expression of the complex modulus in the 2S2P1D model is given in Eq. (4) for a fixed reference  
131 temperature:

$$133 \quad E_{2S2P1D}^*(i\omega\tau_E) = E_{00} + \frac{E_0 - E_{00}}{1 + \delta(i\omega\tau_E)^{-k} + (i\omega\tau_E)^{-h} + (i\omega\beta\tau_E)^{-1}} \quad (4)$$

134  
135 where some parameters are already explained, and  $\beta$  is a parameter linked to the dashpot viscosity  
136  $\eta = (E_0 - E_{00})\beta\tau$  when  $\omega \rightarrow 0$ .

137 As already mentioned, this model is particularly suitable for bituminous mixtures or  
138 materials with almost only viscous dissipation and with seven constants can fully describe the  
139 rheological behaviour of HMA mixtures. However, it is possible that, in case of CBTM mixtures,  
140 the 2S2P1D is not able to well represent the material behaviour.

141



**Figure 1** 2S2P1D analogical representation (taken from (Gayte et al., 2016))

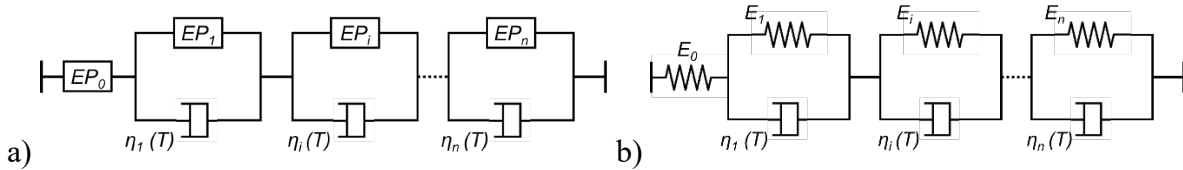
142  
143  
144

145 Analyzing the model parameters in the literature, it is observed that  $E_0$  and  $E_{00}$  are  
146 significantly different between HMA and CBTM mixtures (Carret et al., 2018; Gandi et al., 2017;  
147 Lamothe et al., 2017). In fact, springs stiffness depends mainly on the aggregate skeleton and the  
148 air voids content. In general,  $E_0$  in CBTM mixtures is lower than in HMA due to the higher air  
149 voids and the lower bitumen dosage. On the other hand, the value of  $E_{00}$  in CBTM mixtures is  
150 usually higher than in HMA due to the presence of cement, which constitutes a stiffening phase  
151 also at very high temperatures (or very low frequencies), when the bitumen phase is considered as  
152 fluid. Moreover, the dashpot viscosity and the related parameter  $\beta$  is around 10 times higher in  
153 CBTM than in HMA, indicating the small role played by the dashpot when cement is used with  
154 bitumen.

## 155 2.2 DBN model

156 The DBN model (named after the authors Di Benedetto and Neifar) was specifically proposed to  
157 introduce non-linearity phenomena and to describe large-strain plastic behaviour of granular soils  
158 (Blanc et al., 2011; Di Benedetto et al., 2014). Later, its application was extended to analyze plastic  
159 dissipation phenomena in bituminous mixtures (Di Benedetto, Mondher, et al., 2007; Gayte, 2016;  
160 Neifar & Di Benedetto, 2001). In the model, the non-linearity is represented by means of  
161 elastoplastic (EP) bodies in series with viscous dashpots, the latter representing the time-  
162 temperature dependency. The combination is then repeated  $n$  times ( $n$  elements) in order to  
163 increase the model precision (Figure 2a). When the strain level tends to 0, the DBN model takes  
164 its asymptotic form as a Generalized Kelvin-Voigt (GKV), and the EP bodies are replaced by  
165 springs (Figure 2b). When the number of elements tends to the infinite, the representation passes  
166 from a discrete spectrum to a continuous spectrum. Thanks to the high versatility, this model is  
167 able to describe behaviour for a wide range of solicitations, temperatures and cycle numbers.

168 The EP bodies, which represent a non-viscous behaviour, are generally adopted for non-  
 169 cohesive (or elastoplastic) granular materials (Ashmawy et al., 1995; Blanc et al., 2011; Di  
 170 Benedetto et al., 2014; Tatsuoka et al., 2008). Plastic dissipation can be observed in sands for  
 171 cycles at small strain amplitude, characterized by a hysteretic stabilized behaviour. Considering  
 172 materials composed of aggregates and bituminous binder, the non-viscous behaviour should be  
 173 attributed to aggregates (or RAP in this case), whereas the viscous contribution to the binder. With  
 174 such considerations, it can be assessed that the non-viscous behaviour is independent from  
 175 temperature (and frequency, if the Time-Temperature Superposition Principle stands), which is  
 176 instead affecting the purely viscous part. In case the DBN model is applied to represent plasticity  
 177 phenomena at small cycles number (and small strain domain), it can take a simplified form (Attia,  
 178 2020).  
 179



180 a) DBN model for bituminous mixtures; b) Generalized Kelvin-Voigt model, which  
 181 gives an asymptotic representation of the DBN model when strain tends to 0  
 182  
 183

184 The cyclic response of EP bodies is characterized by a function linking the strain and the  
 185 stress, called “virgin curve” (Figure 3). One property of this function is that the unloading (or  
 186 loading) curve joins tangentially the virgin unloading (or loading) curve at the inverse value of the  
 187 reversal stress (Di Benedetto, Mondher, et al., 2007).

188 For many construction materials, including metals, concrete and soils, the dissipation  
 189 behavior may be expressed by the specific damping capacity  $\psi$ , which, for the KV body and small  
 190 dissipation energy, is calculated as follows (Genta, 2009):  
 191

$$192 \quad \psi = \frac{\Delta W_{LVE}}{W_E} = \frac{\pi \varepsilon_0 \sigma_0 \sin \phi}{1/2 \varepsilon_0 \sigma_0} = 2\pi \sin \phi \quad (5)$$

193  
 194 where  $\Delta W_{LVE}$  is the area of the hysteresis loop with elliptical shape (energy dissipated at each  
 195 cycle),  $W_E$  is the energy stored by the spring at each cycle,  $\varepsilon_0$  is the amplitude of the sinusoidal

196 strain,  $\sigma_0$  is the amplitude of the sinusoidal stress and  $\varphi$  is the frequency-dependent phase angle  
 197 describing the lag between stress and strain in the linear viscoelastic response.

198 The EP bodies store elastic energy and dissipate through time-temperature independent  
 199 mechanisms. As a result, a sinusoidal loading is represented by an hysteresis loop as in Figure 3.

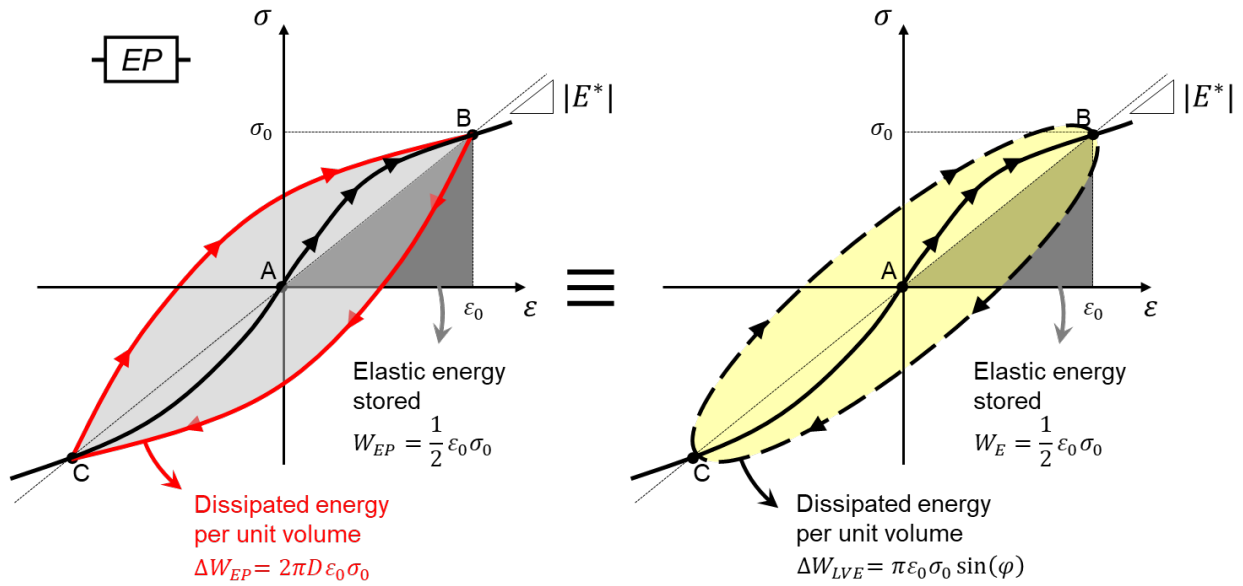
200 The energy  $\Delta W_{EP}$  dissipated by the EP body is computed as:

201

$$202 \Delta W_{EP} = W_{EP}\psi = 1/2 \varepsilon_0 \sigma_0 \psi = 2\pi D \varepsilon_0 \sigma_0 \quad (6)$$

203

204 where  $D = \psi/4\pi$  is an adimensional time-temperature independent damping ratio.



205

206 **Figure 3** Cyclic loading behaviour of elastoplastic element for small number of cycles (path  
 207 ABCB)

208

209 In case the number of cycles applied and the deformation are small, the plastic energy  
 210 dissipation  $\Delta W_{EP}$  can be expressed as an equivalent linear viscoelastic dissipation  $\Delta W_{LVE}$  through  
 211 the definition of an equivalent phase angle, by fixing  $\Delta W_{EP} = \Delta W_{LVE}$  (Figure 3):

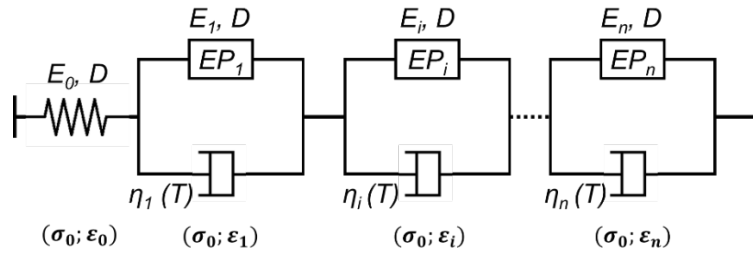
212

$$213 \sin(\varphi_{EP}) = 2D \quad (7)$$

214

215 The version of the DBN model presented in this paper is obtained by the series arrangement  
 216 of units consisting of a viscous and temperature-dependent dashpot  $\eta_i(T)$  in parallel with a EP<sub>i</sub>

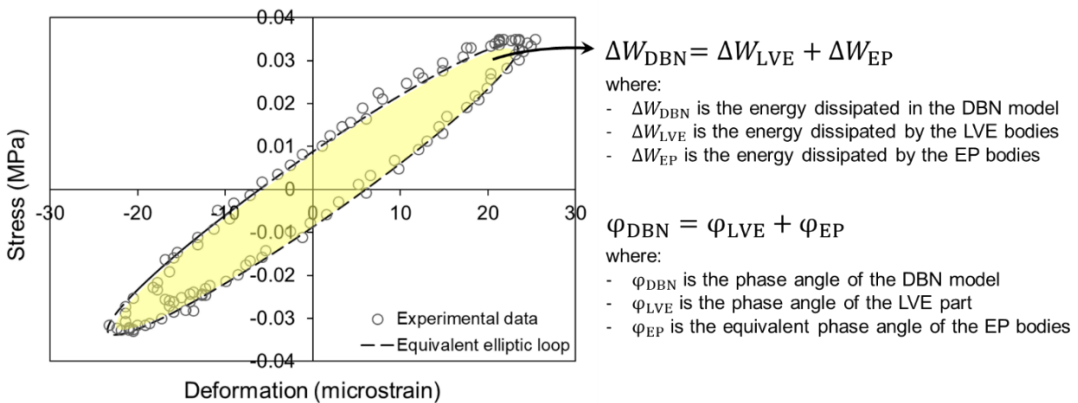
217 body ( $E_i, D_i$ ). Moreover, the dashpot is not present in the first unit ( $\eta_0 = 0$ ) and all units are  
 218 characterized by the same dissipation parameter ( $D_i = D$ ) (Figure 4).



219  
 220 **Figure 4** Representation of the DBN model applied in the small strain domain, EP are  
 221 represented by a spring (modulus, E) and a non-viscous dissipation (D)  
 222 As a consequence, the DBN model phase angle  $\varphi_{DBN}$  is expressed as the total contribution by the  
 223 viscous and non-viscous components (Figure 5):

224  
 225 
$$\sin(\varphi_{DBN}) = \sin(\varphi_{LVE} + \varphi_{EP}) = \frac{\Delta W_{DBN}}{\pi \varepsilon_0 \sigma_0} =$$
  
 226 
$$= \frac{\Delta W_{LVE} + \Delta W_{EP}}{\pi \varepsilon_0 \sigma_0} = \sum_{i=0}^n \left( \frac{\omega \eta_i}{E_i^2 + (\omega \eta_i)^2} + \frac{2 \cdot D \cdot E_i}{E_i^2 + (\omega \eta_i)^2} \right) \cdot |E^*| \quad (8)$$

227  
 228 where  $\Delta W_{DBN}$  is the total cycle dissipation (viscous and non-viscous),  $E_i$  and  $\eta_i$  are the Young's  
 229 modulus and the Newtonian viscosity of the  $i^{th}$  element, respectively,  $\omega$  is the pulsation,  $\varphi_{LVE}$  is  
 230 the phase angle of the viscous dashpot,  $\varphi_{EP}$  is the phase angle of the non-viscous damping, and  
 231 D is the damping ratio (Ashmawy et al., 1995).



233  
 234 **Figure 5** Energy dissipation during cyclic loading in the small strain domain (experimental data  
 235 superposed to an equivalent sinusoidal loading)

236 It is assumed that the addition of non-viscous dissipation does not influence the value of  
 237 the complex modulus, which can be expressed from the GKV configuration:

238

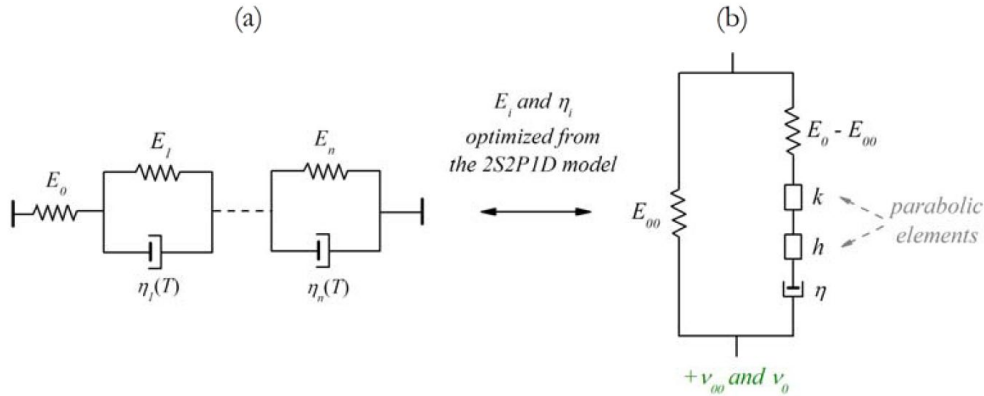
$$239 \quad E_{GKV}^*(i\omega, T) = \left( \frac{1}{E_0} + \sum_{i=1}^n \frac{1}{E_i + i\omega\eta_i(T)} \right)^{-1} \quad (9)$$

240

241 where  $i$ ,  $\omega$  and  $T$  were previously explained,  $E_0$  is the Young's modulus of the first element,  
 242  $E_i$  and  $\eta_i$  were previously explained. The number of elements  $n$  can be chosen arbitrarily to reduce  
 243 the distance between the discrete GKV configuration and the 2S2P1D. In particular, the 2S2P1D  
 244 model should be initially fitted on the norm of the complex modulus of the material, and then the  
 245 GKV model is calibrated according to the 2S2P1D (Figure 6).

246 Consequently for any chosen number of elements ( $n$ ), the DBN model only needs seven  
 247 constants from 2S2P1D plus an additional parameter ( $\varphi_{EP}$ ) to take into account plasticity at small  
 248 strain levels.

249



250

251 **Figure 6** Correlation between: a) GKV model, and b) 2S2P1D model (taken from (Di Benedetto,  
 252 Delaporte, et al., 2007))

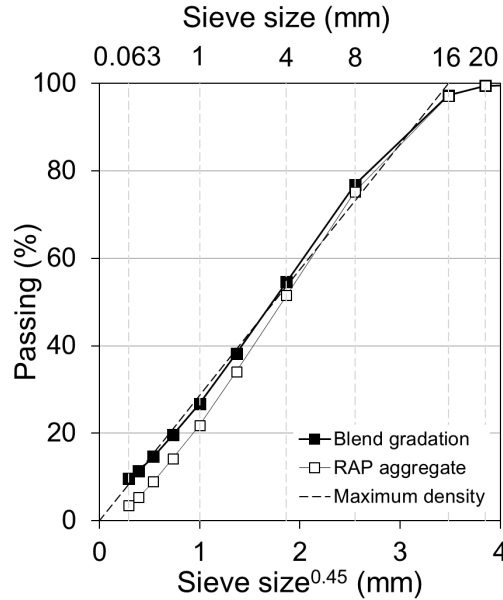
253

### 254 3 MATERIALS AND METHODOLOGY

#### 255 3.1 Materials and mixtures

256 The CBTM mixtures produced for this study are characterized by an aggregate distribution  
 257 composed of 94% of RAP and 6% of limestone filler. The correction with filler allows having a  
 258 gradation close to the maximum density curve (Figure 7). The properties of the RAP aggregate are

259 listed in Table 1. The ordinary Portland cement dosage was fixed at 1.5% by mass of dry  
 260 aggregates. The cement was a GU type (standard CSA A3000) with compressive strength at 28  
 261 days of 43.9 MPa (standard ASTM C109).



262  
 263 **Figure 7** Maximum density curve, RAP and blend gradations

264  
 265 **Table 1** RAP aggregate properties

266

Property	Standard	Unit	Value
Binder content	ASTM D6307	%	5.5
Nominal maximum particle dimension	ASTM D448-03	mm	16
Maximum specific gravity	ASTM C127-128	-	2.482
Average bulk density	LC 21-065, -066 and -067	-	2.323
Water absorption	ASTM C127 and C128	%	1.1

267  
 268 **Table 2** Bitumen emulsions properties

269

Bitumen emulsion properties	Standard	Unit	Emu. A	Emu. B
Density	ASTM D6397-16	g/cm <sup>3</sup>	1.0	n.d.*
Viscosity @ 40 °C	EN 13302	s	N/A	42.5
Residual bitumen content	EN 1428 or ASTM D6997-12	%	60.3	60.0
Storage stability @ 24 hours	ASTM D6930-10	%	0.6	n.d.*
Breaking Index	EN 13075	%	n.d.*	2
<b>Residual bitumen properties</b>				
Penetration @ 25 °C	EN 1426 or ASTM D5-13	mm	4.1	10.0
Softening point	EN 1427 or ASTM D36-14	°C	48.6	43.0

\*n.d.: not determined

270

271 Mixtures were produced with two bitumen emulsions in order to compare two different  
272 bitumen sources. The main properties of both emulsions are listed in Table 2, and they are named  
273 from now on as Emulsion A and Emulsion B.

274 In both cases, the bitumen emulsion dosage was fixed at 5% (3% of residual bitumen) by  
275 mass of aggregates. The total water dosage was fixed at 4.0% by mass of aggregate, in order to  
276 reach the target air voids of 15% without employing high compaction energy and to avoid any  
277 material loss (water, bitumen and/or fine particles) during compaction.

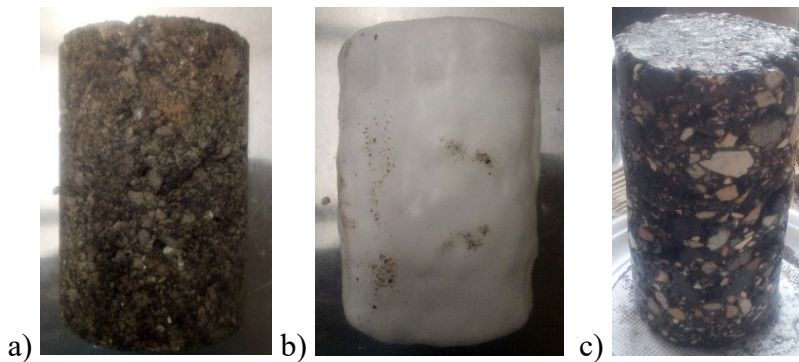
### 278 3.2 Mixtures production

279 After mixing, the specimens were compacted by means of a gyratory compactor (GC) with mould  
280 diameter of 100 mm, constant pressure of 600 kPa, gyrations rate of 30 rpm and internal angle of  
281 1.16°. The volumetric composition of the specimen is monitored with compaction, which is  
282 performed at fixed height to obtain the target value of voids in the mixture ( $V_m$ ) of 15% ± 1%  
283 (Grilli et al., 2016):

$$284 \quad V_m = \frac{V_{V,A} + V_{W,I}}{V} \cdot 100 = \frac{V - (V_S + V_C + V_{B,R})}{V} \cdot 100 \quad (10)$$

287 where  $V$  is the total volume of the specimen,  $V_S$  is the bulk volume of aggregates (in  
288 saturated surface dried condition),  $V_C$  is the volume of cement,  $V_{B,R}$  is the volume of residual  
289 bitumen from emulsion,  $V_{W,I}$  is the volume of intergranular water and  $V_{V,A}$  is the volume of air. A  
290 total of nineteen (19) specimens were produced, but only three (3) are considered in this study to  
291 focus the attention on the proposed model and its description.

292



293 a) un-sealed condition, b) sealed condition and,  
294 **Figure 8** SGC specimen of Ø100 mm x 140 mm: a) un-sealed condition, b) sealed condition and,  
295 c) coring and sawing to obtain the testing specimens of Ø75 x 120 mm

296

297 After compaction, the specimens followed a curing process as shown in Table 3 (Raschia  
 298 et al., 2020). The period lasted 1 year, simulating a long-term curing to reach a quite stable  
 299 condition of the physical and mechanical properties. The first and second curing periods, for a total  
 300 of 28 days, was same for the three specimens. After that, two specimens were kept in unsealed  
 301 conditions for the third curing period, whereas one specimen was wrapped in plastic foil and sealed  
 302 with several layers of wax for a final coating thickness of around 5 mm. The sealed condition after  
 303 28 days was chosen to stop the curing and/or ageing of the material, which instead was promoted  
 304 in the unsealed specimens. At the end of the third curing period, the three specimens were cored  
 305 at a diameter of 75 mm and prepared for complex modulus testing (Figure 8).

306

**Table 3** Mixtures naming and curing process

307

Emulsion	Mixture	Curing types		
		1 <sup>st</sup>	2 <sup>nd</sup>	3 <sup>rd</sup>
		14 days	14 days	11 months
A	A Unsealed	25 °C	40 °C <sup>(1)</sup>	Room temperature (Unsealed)
	A Sealed			Room temperature (Sealed)
B	B Unsealed		40 °C	Room temperature (Unsealed)

<sup>(1)</sup> The curing was performed with controlled relative humidity at  $55 \pm 5 \%$

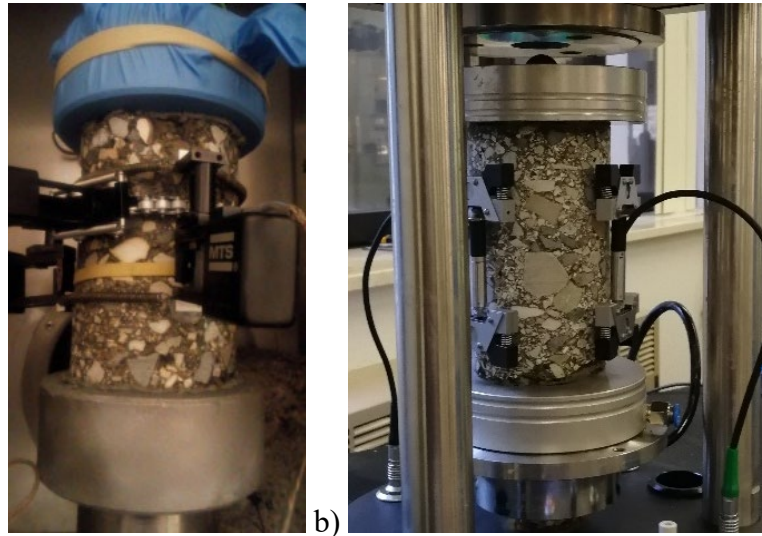
308

### 309 3.3 Experimental devices

310 The experimental program was carried out in two laboratories with different equipments, as a  
 311 collaboration between different institutions. However, it is assumed that testing apparatus does not  
 312 significantly influence results as long as the test is performed in the LVE field and with same  
 313 testing conditions (tension-compression). In case of mixtures with Emulsion A, specimens were  
 314 tested with an MTS press, whereas specimen with Emulsion B was tested with an asphalt mixture  
 315 performance tester pro (AMPT PRO) servo-hydraulic press. However, complex modulus tests  
 316 were performed in both cases in only compression configuration (haversine loading) and the axial  
 317 strain was measured by placing three extensometers in the middle part of the specimen and 120°  
 318 apart (Figure 9). The target axial strain was 50 and 30 microstrain for Emulsion A and Emulsion  
 319 B mixtures, respectively. Specimens with Emulsion A were tested at a temperature range between  
 320 -20 °C and 40 °C with 10 °C steps, while frequencies ranged between 0.1 Hz and 10 Hz. In case

321 of Emulsion B, temperature ranged between 0 °C and 40 °C with 10°C steps, and frequencies  
322 varied between 0.1 Hz and 10 Hz.

323



324

a)

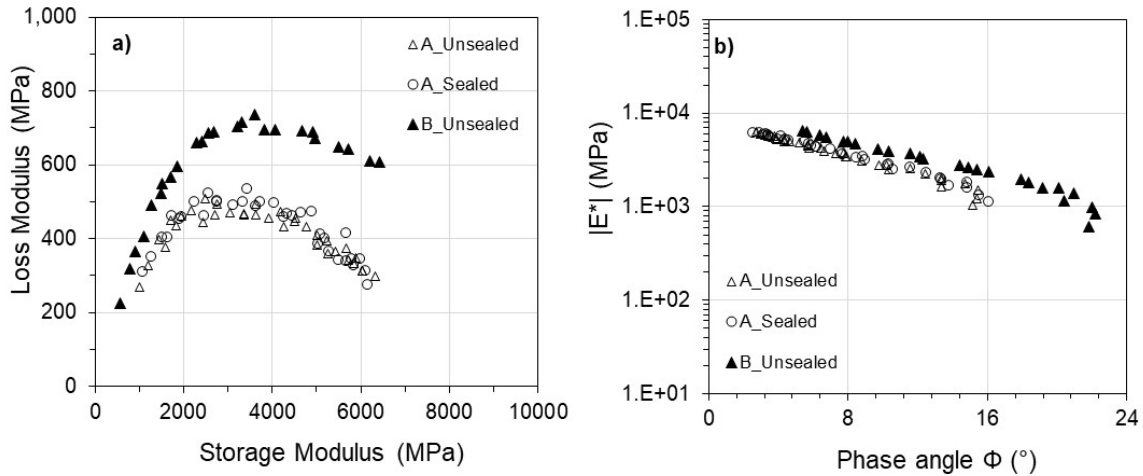
b)

325 **Figure 9** View of a specimen with measurement system in: a) MTS press and, b) AMPT PRO  
326 press

#### 327 4 RESULTS ANALYSIS

328 Figure 10 shows results from the tested mixtures in the Cole-Cole plan and Black space. It  
329 can be observed that in all the cases the experimental points follow a continuous line, indicating  
330 that the Time-Temperature Superposition Principle (TTSP) is respected and the rheological models  
331 described above can be applied. The range of  $|E^*|$  values is quite the same for the three mixtures  
332 studied indicating that the emulsion type and the type of curing did not significantly affect the  
333 stiffness of the mixtures (Figure 10b). On the contrary, it can be observed that mixture B\_Unsealed  
334 is characterized by a different trend of the phase angle when compared to both mixtures with  
335 Emulsion A (Figure 10b). It is reasonable to expect that changing the emulsion, and hence the  
336 residual binder, the viscous properties could have been affected. Moreover, comparing the two  
337 mixtures, A\_Unsealed and A\_Sealed, it is noted that the experimental points are superposed and a  
338 distinction is not possible. Therefore, the sealing condition during the third stage of curing did not  
339 have a clear effect on mixtures produced with Emulsion A, since the material properties did not  
340 change as expected (no further curing and apparently no ageing).

341



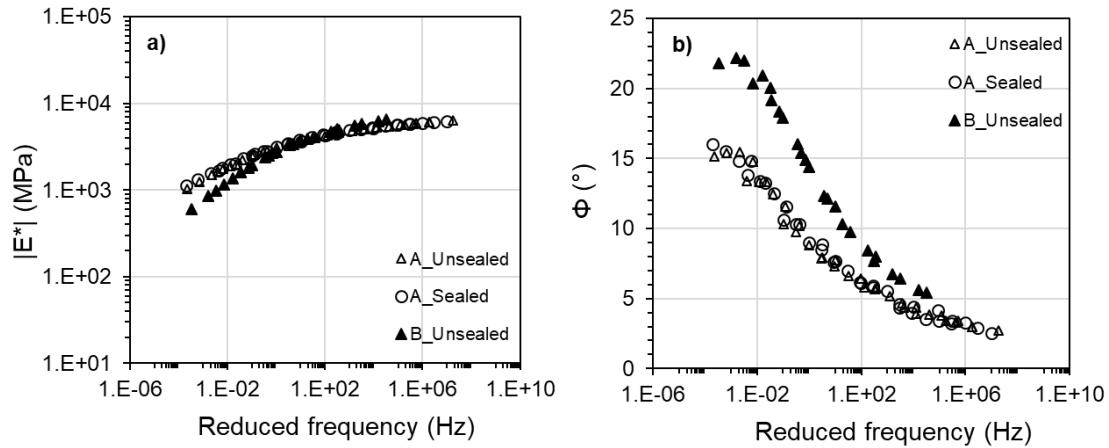
342 **Figure 10** Experimental results showed in: a) Cole Cole plan and, b) Black space  
 343  
 344

#### 345 4.1 Time-Temperature Superposition Principle, TTSP

346 The experimental data show that the TTSP is applicable to CBTM mixtures. As a consequence,  
 347 the isothermal curves of the norm of the complex modulus,  $|E^*|$ , and of the phase angle,  $\phi$ , can  
 348 be shifted in order to obtain the respective master curves (Figure 11).

349 Figure 11 shows the master curves of the norm of the complex modulus and phase angle at  
 350 a reference temperature  $T_{ref} = 20\text{ }^\circ\text{C}$ . As in the previous representation, the effect of the emulsion  
 351 is highlighted on the mechanical properties of mixtures studied. In particular, mixture B\_Unsealed  
 352 showed lower modulus at low frequencies (or high temperatures) and higher modulus at high  
 353 frequencies (or low temperatures), confirming the crucial role of the bituminous binder used in the  
 354 thermal sensitivity of the mixture. Such effect is also visible in the master curve of the phase angle,  
 355 which is globally higher for mixture with Emulsion B compared to mixtures with Emulsion A  
 356 (Figure 11b). Considering that the dosage of residual bitumen is the same, this difference could be  
 357 explained by the fact that the bitumen from Emulsion B is more time-temperature dependant than  
 358 bitumen from Emulsion A.

359

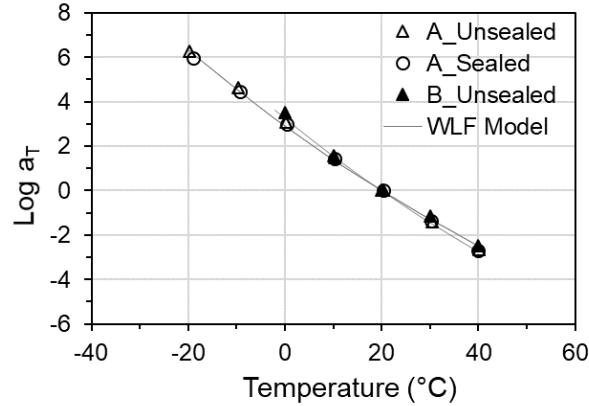


360  
 361 **Figure 11** At  $T_{ref} = 20 \text{ }^{\circ}\text{C}$ , master curves of: a) The norm of the complex modulus  $|E^*|$  and,  
 362 b) The phase angle  $\phi$   
 363

364 Figure 12 shows the shift factors in function of the temperature. The experimental points  
 365 are modelled by the Williams–Landel–Ferry (WLF) model, mathematically expressed as:  
 366

$$367 \log(a_T) = -\frac{C_1(T-T_S)}{C_2+T-T_S} \quad (11)$$

368  
 369 where  $a_T$  is the shift factor,  $C_1$  and  $C_2$  are constants,  $T$  is the temperature and  $T_S$  is the reference  
 370 temperature (Ferry, 1980). It is observed that the shift factors of the three mixtures are significantly  
 371 close at all the temperatures tested (Table 4). The obtained values of  $a_T$  can be compared to the  
 372 ones obtained for HMA mixtures in the literature, tested in the same range of temperatures (Di  
 373 Benedetto et al., 2004). Moreover, such dependency on temperature highlights the thermo-  
 374 mechanical response of CBTM mixtures. It has been shown that in HMA the  $a_T$  coefficients are  
 375 very close between the binder and the related mixture (Di Benedetto et al., 2004). Hence, assuming  
 376 this is also valid for CBTM, it would be possible to have the same shift factors for both residual  
 377 bitumen of the emulsions used.



378  
379

**Figure 12** Shift factors,  $a_T$ , and WLF model related to the studied mixtures

380

381  
382

**Table 4** WLF parameters

Parameters	A Unsealed	A Sealed	B Unsealed
$C_1$	34.7	34.7	38.7
$C_2$	258.5	258.5	258.2

383

#### 384 4.2 The 2S2P1D model

385 Figure 13 shows the experimental data obtained for one mixture (B\_Unsealed) modelled  
386 with the 2S2P1D model. According to the LVE theory for bituminous materials, if the TTSP is  
387 respected, the rheological model should be unique and valid in all the representations: master  
388 curves, Black space and Cole-Cole plan. In Figure 13, three calibrations of the 2S2P1D model are  
389 presented:

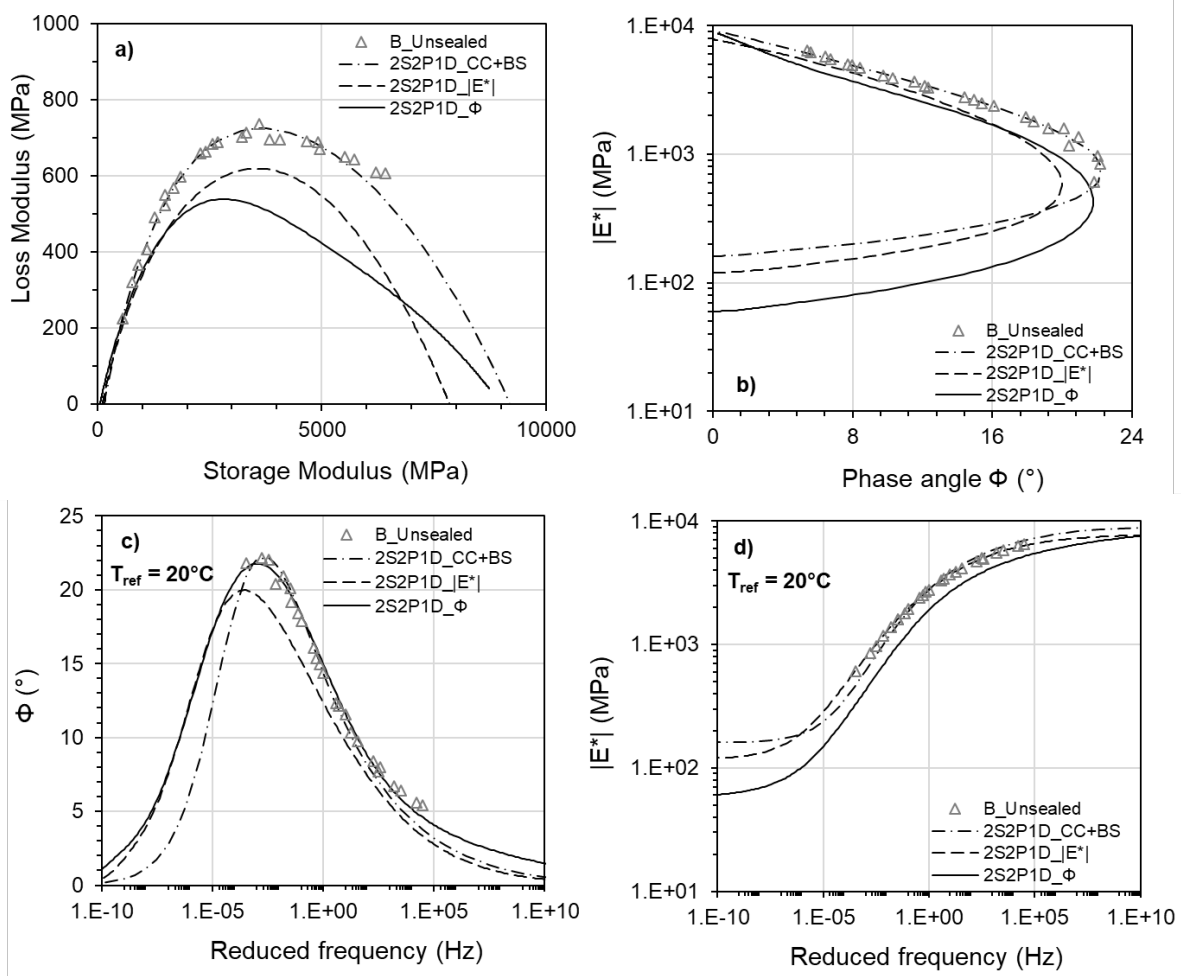
- 390 • optimization from the data in the Cole-Cole and Black spaces, minimizing the error  $|\Delta E^*|$  as  
391 in Eq. (12) (named: 2S2P1D\_CC+BS);
- 392 • optimization from the data plotted in the master curve of the  $|E^*|$ , minimizing the error  
393 dev  $|E^*|$  as in Eq. (13) (named: 2S2P1D\_ $|E^*|$ );
- 394 • optimization from the master curve of  $\phi$  minimizing the error  $\Delta\phi$  as in Eq. (14) (named:  
395 2S2P1D\_ $\phi$ ).

$$396 \quad |\Delta E^*| = \sqrt{(E_{1,\text{exp}} - E_{1,2S2P1D})^2 + (E_{2,\text{exp}} - E_{2,2S2P1D})^2} \quad (12)$$

$$397 \quad \text{dev } |E^*| = \frac{|E^*|_{\text{exp}} - |E^*|_{2S2P1D}}{|E^*|_{\text{exp}}} \cdot 100 \quad (13)$$

398  $\Delta\phi = \phi_{\text{exp}} - \phi_{2\text{S2P1D}}$  (14)

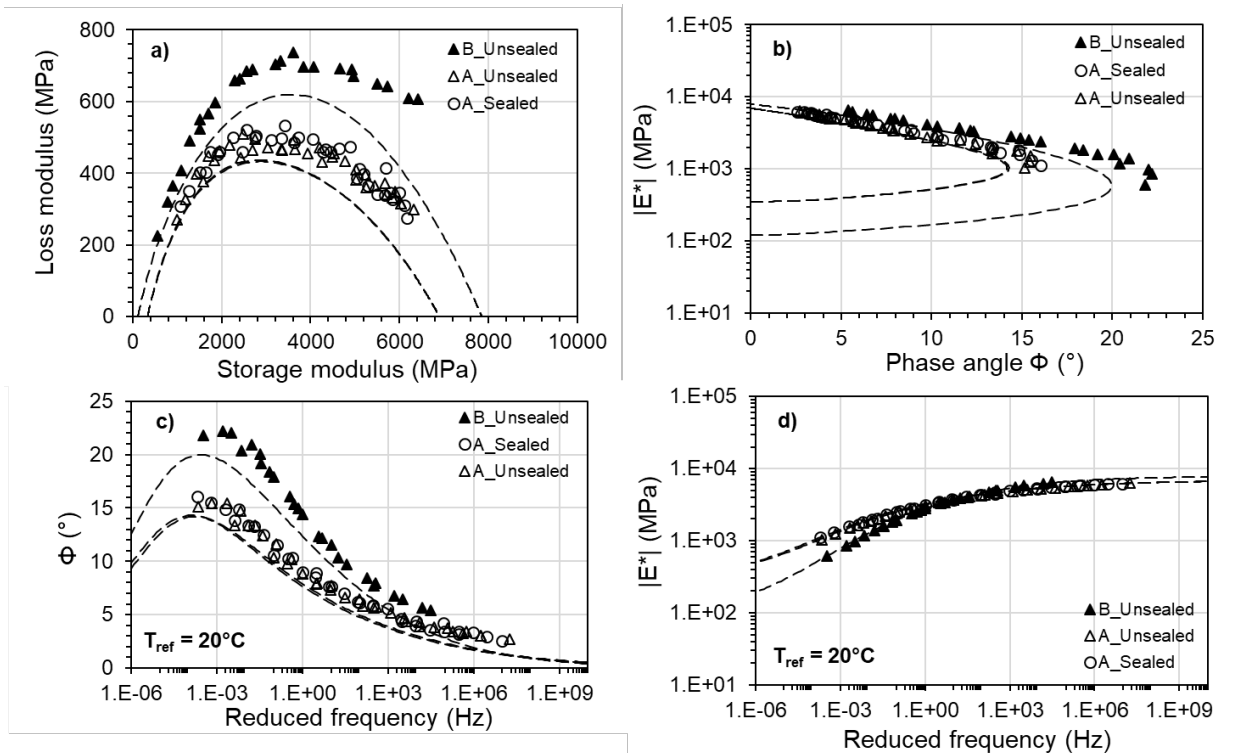
399 where  $E_{1,\text{exp}}$  and  $E_{1,2\text{S2P1D}}$  are the storage moduli of the experimental results and the 2S2P1D  
 400 model, respectively;  $E_{2,\text{exp}}$  and  $E_{2,2\text{S2P1D}}$  are the loss moduli of the experimental results and the  
 401 2S2P1D model, respectively;  $|E^*|_{\text{exp}}$  and  $\phi_{\text{exp}}$  are the experimental results of the norm of the  
 402 complex modulus and the phase angle, respectively;  $|E^*|_{2\text{S2P1D}}$  and  $\phi_{2\text{S2P1D}}$  are the 2S2P1D  
 403 values of the norm of the complex modulus and the phase angle, respectively.



404  
 405 **Figure 13** Optimization of the 2S2P1D model for B\_Unsealed mixture according to: a) Cole-  
 406 Cole plan, b) Black space, c) phase angle master curve and, d) complex modulus master curve  
 407 (T<sub>ref</sub> = 20 °C)  
 408

409 In Figure 13, it is observed that none of the three optimizations superpose among them and  
 410 with the experimental data in all the four representations. The calibration 2S2P1D\_CC+BS does  
 411 not well represent the master curves trend at both low and high reduced frequencies (high and low

412 temperatures, Figure 13c-d). Furthermore, the optimization 2S2P1D\_Φ done on the phase angle  
 413 master curve significantly underestimates the norm of the complex modulus,  $|E^*|$ , visible in the  
 414 three other representations (Figure 13a-b-d). However, the model 2S2P1D\_ $|E^*|$  calibrated on the  
 415 master curve of the  $|E^*|$  underestimates the  $\phi$  of a constant value on the full frequencies range  
 416 (around 2°, Figure 13c). This result can be justified by the presence of a non-viscous dissipation  
 417 which cannot be taken into account with a LVE rheological model such as 2S2P1D.  
 418 As a consequence, the results from the three mixtures studied are modelled fitting the 2S2P1D on  
 419 the norm of the complex modulus (calibration : 2S2P1D\_ $|E^*|$ ) (Figure 14). It is observed that the  
 420 phase angle master curve is not well represented for all the mixtures and this is visible also in  
 421 Black space and Cole-Cole plan (Figure 14a-b). For this reason, the DBN model should be applied  
 422 to consider also the non-viscous contribution in the complex behaviour. The shifting was done by  
 423 means of a closed-form shifting (CFS) algorithm which minimizes the area between two  
 424 successive isothermal curves of  $|E^*|$  and estimates the shift factors (Gergesova et al., 2011).  
 425

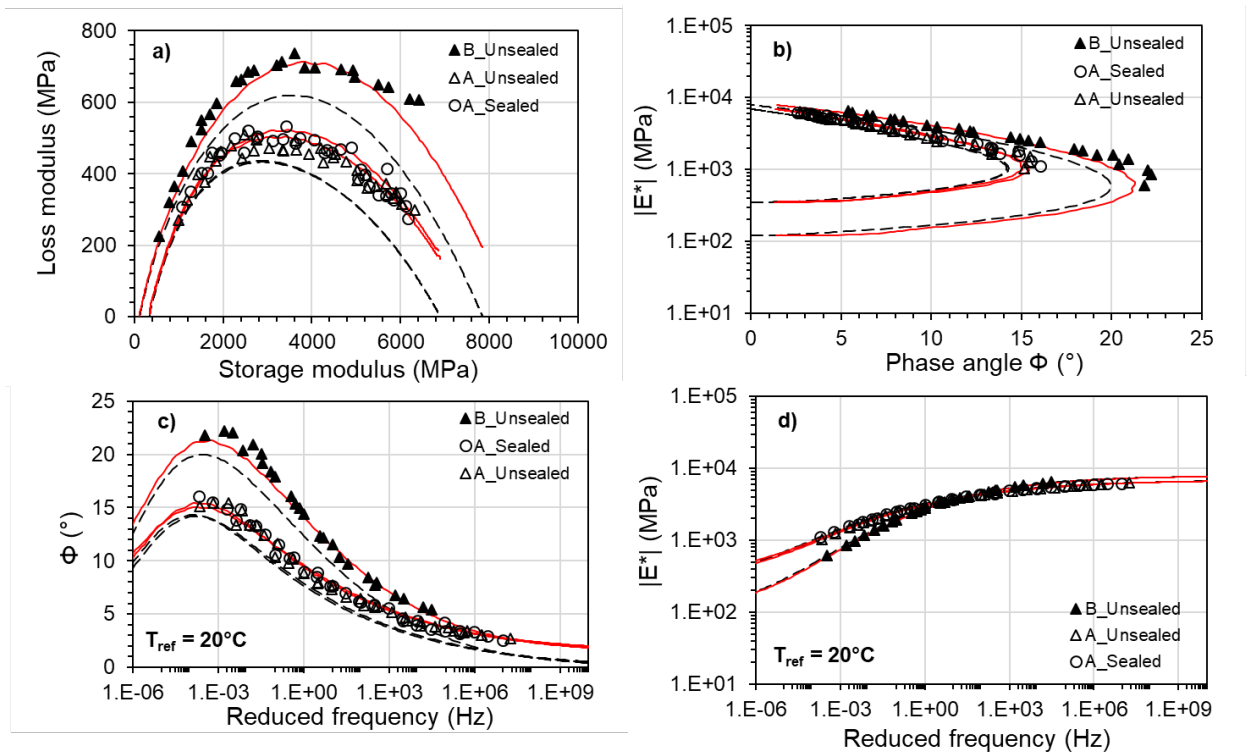


426  
 427 **Figure 14** Application of 2S2P1D model fitted on the master curve of the norm of the complex  
 428 modulus (calibration : 2S2P1D\_ $|E^*|$ ) for: a) Cole-Cole plan, b) Black space, c) phase angle  
 429 master curve and, d) complex modulus master curve ( $T_{ref} = 20\text{ }^{\circ}\text{C}$ )

430 **4.3 The DBN model**

431 Figure 15 shows the experimental results of the three studied mixtures modelled with the  
 432 2S2P1D and DBN models. In order to obtain a good level of precision and correlation with the  
 433 2S2P1D, the number of elements in the GKV model was fixed at 40. The values of  $E_i$  and  $\eta_i$  for  
 434 each element of the model are listed in the appendix A (Table A.1). It can be observed that the two  
 435 models are superposed in the plan of the norm of the complex modulus (Figure 15a), whereas in  
 436 the other plans the difference between the two models is due to the introduction of an equivalent  
 437 phase angle representing the non-viscous dissipation,  $\varphi_{EP}$ . This additional parameter is visible as  
 438 a shifting of the model in the Black space and phase angle master curve, and as a rotation in the  
 439 Cole-Cole plan (Figure 15a-b-c).

440



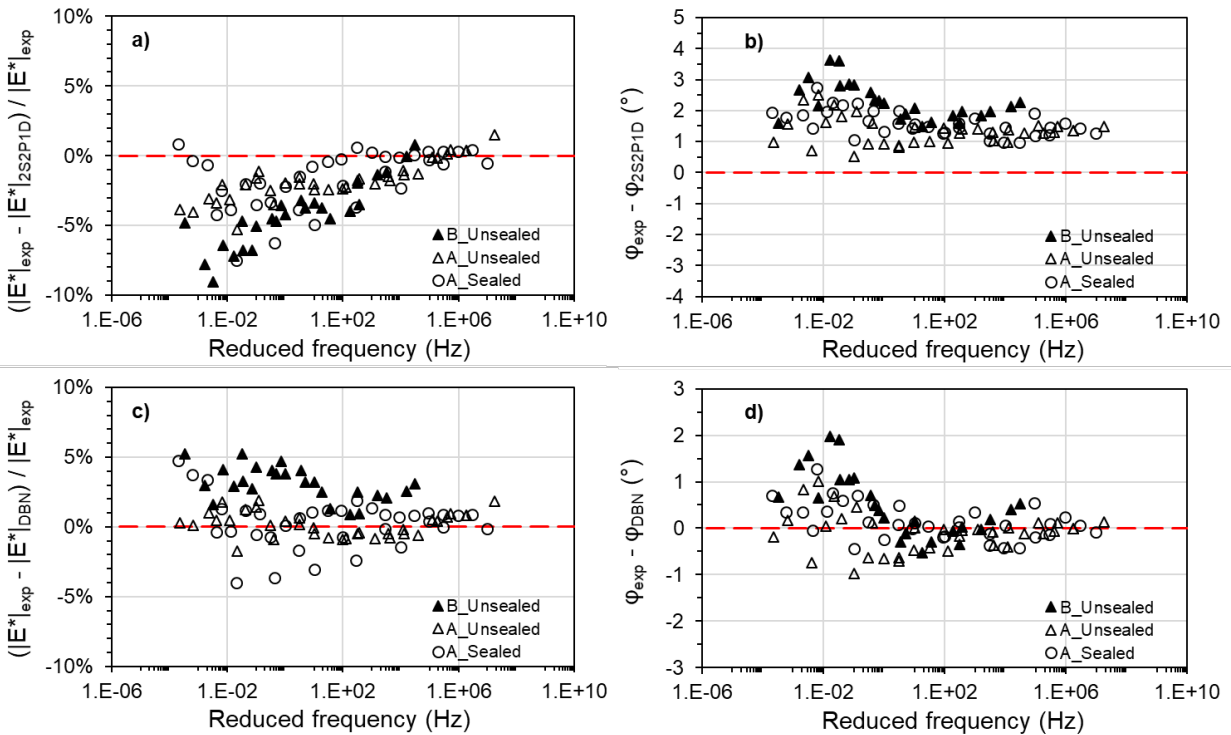
441

442 **Figure 15** Application of the 2S2P1D (dashed line) and DBN (continuous line) models to the  
 443 studied mixtures ( $n = 40$ ): a) Cole-Cole plan, b) Black space, c) phase angle master curve and, d)  
 444 complex modulus master curve ( $T_{ref} = 20\text{ }^{\circ}\text{C}$ )

445

446 **5 DISCUSSION**

447 Figure 16 shows the accuracy of 2S2P1D and DBN models according to the experimental data for  
 448 the whole frequency and temperature ranges. The best fitting in both plans of the DBN model  
 449 compared to the 2S2P1D are highlighted: norm of the complex modulus ( $\pm 5\%$ ) and phase angle  
 450 ( $\pm 2^\circ$ ).



451  
 452 **Figure 16** Accuracy of the model applied: a) 2S2P1D on the norm of complex modulus,  
 453 b) 2S2P1D on the phase angle, c) DBN on the norm of complex modulus, and  
 454 d) DBN on the phase angle

455  
 456 Table 5 lists the model parameters related to the DBN model and the modified version of  
 457 the Huet-Sayegh model (Eq. 3). It is highlighted that the same values can be used by both  
 458 approaches, even if the parameter  $\beta$  needs to be defined in case of DBN model. However, such  
 459 high values make the contribution given by the dashpot almost negligible. This shows that the two  
 460 approaches are able to fit the experimental data adopting the same parameters, but only in the case  
 461 of DBN the material is represented by a rheological model which can be fully applied in the time  
 462 domain and for higher strain rate (out of the LVE field).

463

464  
465

**Table 5** Rheological modelling parameters for the studied mixtures

Mixture	Model	$E_{00}$ (MPa)	$E_0$ (MPa)	$k$ (-)	$h$ (-)	$\delta$ (-)	$\beta$ (-)	$\tau_E$ (20°C) (-)	$\varphi_{EP}$ (°)	$\varphi_{AEP}$ (°)
A_Unsealed	DBN	350	6,900	0.14	0.37	2.25	10E+10	15.0	1.3	-
	HS <sub>q</sub>	350	6,900	0.14	0.37	2.25	-	15.0	-	1.3
A_Sealed	DBN	350	6,900	0.14	0.37	2.25	10E+10	20.0	1.3	-
	HS <sub>q</sub>	350	6,900	0.14	0.37	2.25	-	20.0	-	1.3
B_Unsealed	DBN	120	7,850	0.18	0.39	2.65	10E+05	2.5	1.4	-
	HS <sub>q</sub>	120	7,850	0.18	0.39	2.65	-	2.5	-	1.4

466

467 Comparing parameters for mixes A\_Unsealed and A\_Sealed it is observed that the same  
468 are adopted, meaning that the sealed curing prevented further curing and ageing of the mixtures.  
469 The first two curing periods for a total of 28 days (14 days at 25 °C and 14 days at 40 °C) were  
470 enough to reach a stable condition of the material properties, which were not affected by ageing  
471 as much. The most important factor that affected the rheology of CBTM mixtures was the different  
472 emulsion (i.e. different residual bitumen) used to produce the mixtures. In particular, important  
473 differences could be highlighted comparing model parameters for mixes A\_Unsealed and  
474 B\_Unsealed. Bitumen from Emulsion B conferred to the CBTM mixture a higher modulus value  
475 at high frequencies (or low temperatures) and lower modulus at low frequencies (or high  
476 temperatures), which means a global higher temperature dependency. Moreover, the parameters  
477 related to the viscous part of the model  $k, h, \delta$  are lower for mixture A\_Unsealed, highlighting the  
478 fact that bitumen from Emulsion A gives a less viscous response compared to Emulsion B. The  
479 same conclusion can be confirmed by the values of the characteristic time  $\tau_E$ . Generally, the higher  
480 the value of  $\tau_E$ , the lower is the viscous contribution given by the binder. From these results, the  
481 difference between Emulsion A and B is one order of magnitude.

482 The non-viscous parameter  $\varphi_{EP}$  is almost the same for curing confinement and emulsion  
483 used, meaning that it does not depend on the residual bitumen and confirming that the curing  
484 confinement did not change the rheological response of mixtures. Being a parameter used to  
485 represent frictional or slightly plastic phenomena it is reasonable to assume that it could depend  
486 on the air voids content, bitumen dosage and/or the type of aggregates used. However, it is believed  
487 that the effect of non-viscous dissipation is reversible for a small number of cycles. Since these  
488 aspects were not analyzed in this study, further work is needed to clarify the role of the non-viscous  
489 component in CBTM mixtures.

## 490 6 CONCLUSIONS

491 This paper deals with the thermo-rheological modelling of CBTM mixtures in the small  
492 strain domain. An innovative approach is proposed employing the visco-plastic model DBN  
493 proposed in the literature. The paper focuses on the description of the DBN model application to  
494 CBTM materials; **however, the new approach was applied to preliminarily** study the effects of  
495 curing confinement type and emulsion source in the long-term properties of the CBTM mixtures  
496 studied. The following conclusions can be drawn:

- 497 • DBN is a suitable rheological model to well represent the thermo-rheological behaviour of  
498 cement-bitumen treated materials (CBTM) in the small strain domain. With 8 parameters it  
499 is possible to include in the same model both viscous and non-viscous responses obtaining  
500 an optimal fitting of the experimental results. The equivalent phase angle,  $\varphi_{EP}$ , represents  
501 a non-viscous dissipation parameter typically observed at higher levels of deformation, but  
502 useful in this study to consider frictional and/or plastic phenomena for the CBTM mixtures.  
503 From the results obtained, the  $\varphi_{EP}$  does not seem to depend on binder type and curing  
504 procedure. Using a different model would bring to the definition of different parameters (of  
505 stiffness and dissipation) which could lead to a misunderstanding of the material properties.  
506 This would bring to mistakes if more mixtures are compared (for example effect of air voids,  
507 gradation, bitumen type, etc.). Furthermore, the DBN model can be extended in the time  
508 domain, in order to characterize the material also at higher deformation rates (fatigue).  
509 Additional work is needed to improve the knowledge with regards to such new aspects in  
510 cold materials;
- 511 • Mixtures were cured for 14 days at 25 °C and 14 days at 40 °C in unsealed conditions. After  
512 that, a curing process of 11 months in sealed and unsealed conditions was followed, after  
513 which rheological properties were measured. Results showed that in both conditions the  
514 same stiffness was reached, meaning that the evolution of properties was not **probably**  
515 influenced by sealed or unsealed curing. It **can be** assumed that in sealed condition stiffness  
516 evolution was slowed down or stopped. The same mixture composition was employed to  
517 produce CBTM mixtures with two different emulsion sources, hence different residual  
518 binder. The emulsions chosen are present in the market as specific for cold recycling projects  
519 and they have the same raw characteristics: cationic, slow-setting emulsions with unmodified

520 binder. Nonetheless, results obtained are significantly affected by the type of residual  
 521 bitumen, meaning that it is not an aspect that should be neglected in the mix design.

522 Future studies should focus on improving the application of the DBN model for cold  
 523 materials to enhance the currently missing scientific knowledge of the material. **In particular,**  
 524 **attention should be dedicated to the deeper understanding of the elastoplastic dissipation and the**  
 525 **variables that could affect such property. Moreover, a study focused on the repeatability of the**  
 526 **CBTM complex modulus testing on a larger number of specimens and the following application**  
 527 **of the DBN model would improve the fundamental knowledge of such materials and the suitability**  
 528 **of the model proposed.**

529  
 530  
 531  
 532  
 533

534 **7 APPENDIX A**

535 **Table A.1** Generalized Kelvin-Voigt (GKV) parameters for 40 elements  
 536

Element	A_Unsealed		A_Sealed		B_Unsealed	
	$E_i$	$\eta_i$	$E_i$	$\eta_i$	$E_i$	$\eta_i$
0	6.90E+03	-	6.85E+03	-	7.85E+03	-
1	2.73E+06	1.92E-12	3.35E+06	2.35E-12	8.64E+06	6.08E-12
2	4.28E+06	1.33E-11	5.17E+06	1.61E-11	1.23E+07	3.82E-11
3	2.90E+06	3.99E-11	3.54E+06	4.87E-11	9.43E+06	1.30E-10
4	2.97E+06	1.81E-10	3.76E+06	2.29E-10	1.31E+07	7.95E-10
5	3.86E+06	1.04E-09	4.76E+06	1.28E-09	7.62E+06	2.05E-09
6	2.57E+06	3.06E-09	2.96E+06	3.52E-09	7.00E+06	8.33E-09
7	2.40E+06	1.26E-08	2.88E+06	1.51E-08	4.66E+06	2.45E-08
8	1.75E+06	4.07E-08	2.00E+06	4.66E-08	3.96E+06	9.21E-08
9	1.55E+06	1.59E-07	1.81E+06	1.86E-07	2.79E+06	2.87E-07
10	1.18E+06	5.34E-07	1.33E+06	6.06E-07	2.27E+06	1.03E-06
11	1.01E+06	2.02E-06	1.15E+06	2.32E-06	1.66E+06	3.33E-06
12	7.84E+05	6.97E-06	8.79E+05	7.81E-06	1.31E+06	1.17E-05
13	6.58E+05	2.59E-05	7.40E+05	2.91E-05	9.77E+05	3.84E-05
14	5.21E+05	9.06E-05	5.76E+05	1.00E-04	7.63E+05	1.33E-04
15	4.32E+05	3.32E-04	4.77E+05	3.67E-04	5.73E+05	4.41E-04
16	3.45E+05	1.17E-03	3.76E+05	1.28E-03	4.43E+05	1.51E-03

17	2.84E+05	4.27E-03	3.08E+05	4.63E-03	3.35E+05	5.04E-03
18	2.28E+05	1.52E-02	2.45E+05	1.63E-02	2.57E+05	1.71E-02
19	1.86E+05	5.48E-02	1.99E+05	5.85E-02	1.95E+05	5.72E-02
20	1.50E+05	1.95E-01	1.59E+05	2.06E-01	1.48E+05	1.93E-01
21	1.22E+05	7.00E-01	1.28E+05	7.36E-01	1.12E+05	6.44E-01
22	9.78E+04	2.49E+00	1.02E+05	2.59E+00	8.47E+04	2.15E+00
23	7.88E+04	8.86E+00	8.17E+04	9.17E+00	6.35E+04	7.13E+00
24	6.29E+04	3.12E+01	6.46E+04	3.21E+01	4.74E+04	2.36E+01
25	5.00E+04	1.10E+02	5.12E+04	1.12E+02	3.51E+04	7.71E+01
26	3.92E+04	3.81E+02	4.00E+04	3.89E+02	2.57E+04	2.50E+02
27	3.05E+04	1.31E+03	3.11E+04	1.34E+03	1.86E+04	7.99E+02
28	2.34E+04	4.44E+03	2.39E+04	4.53E+03	1.33E+04	2.52E+03
29	1.77E+04	1.48E+04	1.81E+04	1.52E+04	9.34E+03	7.85E+03
30	1.31E+04	4.87E+04	1.36E+04	5.05E+04	6.47E+03	2.40E+04
31	9.64E+03	1.58E+05	1.01E+04	1.66E+05	4.43E+03	7.27E+04
32	7.05E+03	5.12E+05	7.53E+03	5.47E+05	3.01E+03	2.19E+05
33	5.21E+03	1.67E+06	5.66E+03	1.82E+06	2.06E+03	6.61E+05
34	3.99E+03	5.66E+06	4.39E+03	6.23E+06	1.45E+03	2.06E+06
35	3.27E+03	2.05E+07	3.61E+03	2.26E+07	1.08E+03	6.80E+06
36	2.98E+03	8.28E+07	3.24E+03	9.00E+07	9.05E+02	2.51E+07
37	3.10E+03	3.80E+08	3.27E+03	4.01E+08	8.81E+02	1.08E+08
38	3.62E+03	1.97E+09	3.65E+03	1.98E+09	1.02E+03	5.54E+08
39	4.55E+03	1.09E+10	4.34E+03	1.04E+10	1.37E+03	3.28E+09
40	4.37E+03	4.64E+10	3.64E+03	3.86E+10	9.48E+02	1.01E+10

537

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